EFFECT OF ANNEALING ON MICROSTRUCTURE AND MECHANICAL PROPERTIES OF 0.06% CARBON STEEL

PHOUMIPHON NORDALA

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EFFECT OF ANNEALING ON MICROSTRUCTURE AND MECHANICAL PROPERTIES OF 0.06% CARBON STEEL

by

PHOUMIPHON NORDALA

Thesis submitted in fulfilment of the requirements for the degree of Doctor of Philosophy

DECLARATION

I hereby declare that I have conducted, completed the research work and written the

thesis entitles "Effect of Annealing on Microstructure and Mechanical Properties

of 0.06% Carbon Steel". I also declare that it has not been previously submitted for

the award of any degree of diploma or other similar title of this for any other

examining body or University.

Signature:

Name of Candidate: Phoumiphon Nordala

Matrix Number : P-GD0013/13(R)

Date:

Witness by

Signature:

Main-Supervisor : Assoc. Prof. Ahmad Badri Bin Ismail

Date:

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LIST OF SYMBOLS

% Percentage

%C Percentage of Carbon

%wt Weight Percentage

%EL Percentage of Elongation

F Ferrite

M Martensite

P Pearlite

> More Than

< Less Than

~ Approximately/ About

°C Degree Celsius

K Kelvin

g Gram

Min Minute

MPa Megapascal

mm Milimeter

μm Micrometer

Nano Nanometer

Fe₃C Cementite/Carbide

γ Austenite (gamma)

α Ferrite (alpha)

 λ Wavelength of the Rays

θ Angle between the Incident Rays and the Surface of the Crystal

D Spacing between the Layers of Atoms

T_m Melting Temperature

M_s Martensite Start

 $M_{\rm f}$ Martensite Finish

 ϵ_{eq} Equivalent Strain

n Total Number of Rolling Passes

r Reduction at Each Pass

LIST OF ABBREVIATIONS

DP Dual Phase

UFG Ultrafine Grain

SPD Severe Plastic Deformation

ECAP Equal Channel Angular Pressing

ARB Accumulative Roll-Bonding

HPT High Pressure Torsion

MDF Multidirectional forging

ASTM American Society for Testing and Materials

AISI American Iron and Steel Institute

SAE Society of Automotive Engineers

FCC Face Centered Cubic

BCC Body Centered Cubic

BCT Body Cetered Tetragonal

IT Isothermal Transformation

TWIP Twinning-Induced Plasticity

AC Air-Cooling

IWQ Ice-Water Quenching

SQ Step-Quenching

CR Cold Rolling

OM-IA Optical Microscopy Analysis with attached Image Analyzer

XRD X-ray Diffraction

FESEM Field Emission Scanning Electron Microscopy

EDX Energy Dispersive X-ray

HV Vickers Hardness

RD Rolling Direction

TD Transverse Direction

KESAN PENYEPUHLINDAPAN KE ATAS MIKROSTRUKTUR DAN SIFAT-SIFAT MEKANIKAL BAGI KELULI KARBON 0.06%

ABSTRAK

Kajian ini bertujuan untuk mengkaji mikrostruktur dan sifat mekanik keluli rendah karbon (0.06% C) terhadap beberapa proses rawatan haba dan kerja sejuk. Proses ini bermula dengan tiga rawatan haba yang berbeza; Penyejukan Udara (AC), Pelindapkejutan Air dan Ais (IWQ), dan Penyepuhlindapan Bertingkat (SQ) sebelum 75% pengelekkan-sejuk. Semua spesimen tergelek-sejuk disepuhlindap pada suhu sub-genting dan inter-kritikal. Keputusan menunjukkan bahawa IWQ adalah kaedah terbaik untuk memulakan mikrostruktur awal berbanding dengan kaedah AC dan SQ. Suhu penyepuhlindapan optimum sub-genting adalah pada 525°C. Pada suhu ini (525°C), butiran ferit ultra-halus diperolehi (~ 500 nm). Kekuatan tegangan dan mikro-kekerasan adalah lebih tinggi daripada keadaan diterima kira-kira 89% dan 79% masing-masing. Tambahan pula, didapati bahawa suhu penyepuhlindapan inter-kritikal yang optimum adalah pada 800°C. Pada suhu ini (800°C), kombinasi ferit dan martensit yang terbaik terbentuk. Kekuatan tegangan dan mikro-kekerasan adalah lebih tinggi daripada spesimen yang diterima lebihkurang 81% dan 104%, masing-masing.

EFFECT OF ANNEALING ON MICROSTRUCTURE AND MECHANICAL PROPERTIES OF 0.06% CARBON STEEL

ABSTRACT

The present study is aimed to examine the microstructure and mechanical properties of low-carbon (0.06% C) steel against several heat treatment and cold worked processes. The process was started with three different heat treatments; Air Cooling (AC), Ice-Water Quenching (IWQ), and Step Quenching (SQ) prior 75% cold rolling. All cold-rolled specimens were subcritical and intercritical annealing. The results show that the IWO is the best method to start initial microstructure compared to AC and SQ method. The optimum of subcritical annealing temperature is at 525°C. At this temperature (525°C), ultrafine grains are obtained (~ 500 nm). Tensile strength and microhardness are higher than as-received condition about 89% and 79%, respectively. Furthermore, it was found that the optimum intercritical annealing temperature is at 800°C. At this temperature (800°C), a good combination ferrite and of martensite formed. The tensile strength and microhardness are higher than the as-received specimen about 81% and 104%, respectively.

CHAPTER ONE

INTRODUCTION

1.1 Introduction

In general terms, iron and steel are often confusing. Iron is a chemical element (atomic number 26 on the periodic table). The symbol of chemical 'Fe' is from the Latin word for iron, Ferrum. The word steel is used to describe almost all alloy of iron and it is often mentioned that steel is an alloy of iron and other elements, primarily carbon (Hosford, 2012). The basic material in human development is steel. An adequate quantity of quality steel materials is essential for the realization of industrial development of countries all over the world and for supplying conditions for the modern lifestyle of humanity. Steel is a necessary component of all machinery used for the manufacture of all our goods (Weng, 2009). The study of steels is essential because of the steel represents by far the most extensively-utilized metallic materials, basically due to steel can be manufactured cheaply in large quantities to accurate specifications (Bhadeshia et al., 2011). On the other hand, the wide range of desirable properties which can be managed easily by altering the elemental composition and processing.

The increasing rapidly of motor vehicles worldwide and the associated environmental impact, lightweighting automobile has become an urgent global initiative. In order to provide automobile market requirements such as lightweight materials to improve vehicles with high fuel efficient. The automobile industry has developed steels which are advanced high strength steels (AHSS) together with dual-phase (DP) steels. These steels display both high strength and excellent formability

with weight reduction (Cornette et al., 2001; Shaw et al., 2001). The most common steels used in the automobile industry are mild steels, which are low-carbon steels due to it is cheap and excellent deep draw ability.

Figure 1.1 is a schematic of the Ford Car-2014 model, displaying the dissimilar types of materials that used for designing the body of the car. It can be seen that the bulk of the car body consists of dual-phase steels and mild steels which are low-carbon steels are predominantly used. Hence, the present study is attempting to investigate a plain low-carbon steel.

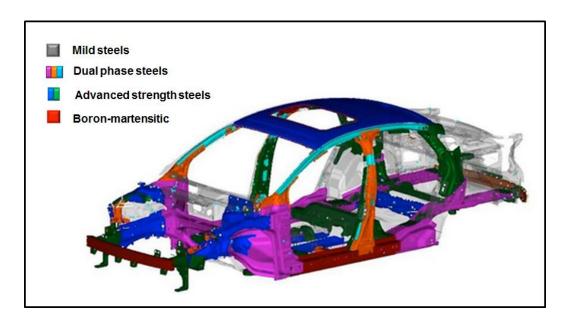


Figure 1.1 Schematic of Ford Car-2014 model, displaying the dissimilar types of materials for designing the body of the car (Mike, 2016).

1.2 Problem statement

Low-carbon steel is known as low mechanical properties and they have very limited applications compare to other higher mechanical properties steels. Many research groups have investigated various innovative methods to improve the strength of low-carbon steel. The methods include Severe Plastic Deformation (SPD) processes such as Equal Channel Angular Pressing (ECAP) (Nejnoto et al., 1998; Horita et al., 2000; Valiev et al., 2006), Accumulative Roll-Bonding (ARB) (Saito et al., 1998; Tsuji et al., 2002b), and High Pressure Torsion (HPT) (Valiev, 1997; Huang et al., 2003). By using these methods, ultrafine grains steel (grains size smaller than 1 µm) with superior mechanical properties (Tsuji et al., 2002b; Valiev et al., 2006) can be achieved.

Besides the SPD processes have been reported by other researchers as well. Fu et al., (2007) had improved the strength of low-carbon (0.10-0.25% C) steels by adding elements such as Titanium (Ti) and Cerium rare earth (Ce) into steels. They had found that tensile strength improved up to about 36% compared with the sample without addition of Ti and Ce. In addition, Li et al., (2014a) lately enhanced mechanical properties of low-carbon (0.091% C) steel by adding Manganese (Mn) and Titanium (Ti) alloy elements. The results showed a yield strength of 920 MPa with elongation about 10%.

Although all these methods have played a big role in improving mechanical properties of low-carbon steels, but they also show drawbacks as following:

(i) SPD processes are difficult for bulk production, particularly in producing large dimension of UFG steel with appropriate strength