SILICA NANOPARTICLES, SYNTHESIS, MODIFICATION AND THEIR APPLICATIONS IN THE FABRICATION OF POLYPYRROLE-BASED SILICA NANOCOMPOSITES

MOHAMMAD JAFARZADEH

UNIVERSITI SAINS MALAYSIA

2010

SILICA NANOPARTICLES, SYNTHESIS, MODIFICATION AND THEIR APPLICATIONS IN THE FABRICATION OF POLYPYRROLE-BASED SILICA NANOCOMPOSITES

by

MOHAMMAD JAFARZADEH

Thesis submitted in fulfillment of the requirements for the degree of Doctor of Philosophy

March 2010

DEDICATION

To my family

And

To my love; Elnaz

ACKNOWLEDGEMENTS

There are many people that I would like to acknowledge for their support along the way. Above all, I wish to express sincere thanks to my main supervisor, Professor Ismail Ab. Rahman, School of Dental Sciences, Health Campus (since Nov. 2008), for his valuable guidance, motivation, patience, and supports throughout the completion of this work both intellectually and financially.

I wish to extend special thanks to my co-supervisor, Associ Prof. Coswald Stephen Sipaut @ Mohd. Nasri, Universiti Malaysia Sabah (since July 2009), for his advice, assistance, encouragement and support during progress of this research.

I would like to acknowledge Universiti Sains Malaysia (USM) and School of Chemical Sciences for the facilities and supports provided during my studies. I would like to express my special thanks to the Institute of Graduate Studies (IPS) for offering a USM-Fellowship scheme to me. I would also like to thank Malaysian Ministry of Higher Education for supplying the financial assistance for this research through FRGS (Fundamental Research Grant Scheme) grant.

I wish to acknowledge all the faculty members and staffs of the School of Chemical Sciences, USM, especially Mr. Ali Zaini, Mr. Burhanudin, Mr. Yee Chin Leng, and others for their assisting over the course of this project. I am also grateful to Mr. Pachamuthu, Ms. Jamilah, Mr. Johari and Ms. Faiza from the Electron Microscopy Unit, USM for their tremendous help in conducting the TEM, SEM and EDX analysis.

I owe my deepest gratitude to all my family members: Hassanjan (father), Fattaneh (mother), Ali (brother) and Hodes (sister in-law) for their encouragement and great financial support, and especially my fiancée, Ms. Elnaz Maleki for her endless love and continuous moral support.

Last, I would like to thank Dr. Ali Parsa, Dr Ramin Tehrani, Mrs Farbia Bakhtiyarizadeh, Ms Soheila Faraji, Mr Mohmmad Etesami, Ms Fatemeh Salehi Karoonian, Mr Ebrahim Akbarzadeh, Ms Atoosa Haghighizadeh, Dr Vejayakumaran Padavettan, Dr. Oo Chuan Wei, Mr Gannty, Mr Daniel Tan, and to all my Iranian friends for being good friends and supportive during my stay in Penang, Malaysia.

Mohammad Jafarzadeh March 2010

TABLE OF CONTENTS

	Page
ACKNOWLEDGEMENTS	ii
TABLE OF CONTENTS	iv
LIST OF TABLES	xii
LIST OF FIGURES	xiv
LIST OF ABBREVIATIONS	XX
ABSTRAK	xxii
ABSTRACT	xxiv

CHAPTER 1: INTRODUCTION

1.1	Nanop	particles	1
1.2	Silica	nanoparticles	1
	1.2.1	Sol-gel technique	2
	1.2.2	Applications of nanoilica	4
1.3	Modif	ication process	5
	1.3.1	Post-modification (grafting)	5
	1.3.2	In-situ modification (co-condensation method)	6
	1.3.3	Incorporation modification	7
1.4	Nanoc	composites	7
	1.4.1	Preparation methods of nanocomposites	9
		1.4.1.1 Sol-gel process for nanocomposites preparation	9
		1.4.1.2 In-situ polymerization process for nanocomposites	
		Preparation	9
		1.4.1.3 Blending process for nanocomposites preparation	10
	1.4.2	Core-shell nanocomposites	10
		1.4.2.1 Preparation methods for core-shell nanocomposites	11
		1.4.2.2 Application of core-shell nanocomposites	11
1.5	Conducting polymers		
	1.5.1	Polypyrrole	12
	1.5.2	Preparation methods of polypyrrole	13

	1.5.3 Application of conducting polymers	14
	1.5.4 Conducting polymer-inorganic nanoparticle composites	14
1.6	Problem statements	15
1.7	Research objectives	17
1.8	Scope of the study	18
1.9	Thesis layout	19

REFERENCES

20

CHAPTER 2: LITERATURE REVIEWS

2.1	Synthe	esis of silica nanoparticles	28
	2.1.1	Mechanism of nucleation and growth of silica nanoparticles	28
	2.1.2	Emulsion route for the preparation of silica nanoparticles	32
	2.1.3	Sol-gel route for the preparation of silica nanoparticles	33
		2.1.3.1 The effect of solvent on particle size	34
		2.1.3.2 The effect of catalyst	36
		2.1.3.3 Effect of drying techniques on morphology	37
	2.1.4	Other effective parameters on sol-gel preparation of	
		silica nanoparticles	39
2.2	Modif	ication of silica nanoparticles	43
	2.2.1	Post-modification of silica nanoparticles (Grafting)	45
	2.2.2	In-situ modification of silica	48
	2.2.3	The optical properties of silica and modified silica	50
2.3	Synth	esis of polypyrrole-based silica nanocomposites	52
	2.3.1	Chemical preparation of polypyrrole	54
	2.3.2	Preparation of polypyrrole-silica nanocomposites	59
	2.3.3	Preparation of polypyrrole-modified silica nanocomposites	65

REFERENCES

68

CHAPTER 3: MATERIALS AND METHODS

3.1	Chem	icals		77
3.2	Equip	ments		77
3.3	Exper	imental pr	ocedures: Part 1	77
	3.3.1	General	and standard procedure for the preparation of	
		silica na	noparticles	77
	3.3.2	Synthesi	s of nanosilica at different feeding rates of ammonia	78
	3.3.3	Synthesi	s of nanosilica at different mixing modes of the reactants	78
		3.3.3.1	Synthesis of nanosilica particle according to Mode A	78
		3.3.3.2	Synthesis of nanosilica particle according to Mode B	79
		3.3.3.3	Synthesis of nanosilica particle according to Mode C	79
	3.3.4	Synthesi	s of nanosilica at different drying processes	80
	3.3.5	Determin	nation of silanol concentration	80
3.4	Exper	imental pr	ocedures: Part 2	80
	3.4.1	General	and standard procedure for the preparation of modified	
		silica na	noparticles via co-condensation method	80
	3.4.2	Preparat	ion of modified silica nanoparticles with different loading	
		via co-co	ondensation method	81
	3.4.3	Preparat	ion of modified silica nanoparticles in the presence	
		of metha	nol via co-condensation method	82
	3.4.4	Preparat	ion of modified silica nanoparticles via post-	
		modifica	ation method with APTES in non-aqueous media	82
	3.4.5	Preparat	ion of modified silica nanoparticles via post-	
		modifica	ation method with APTES in aqueous media	83
	3.4.6	Preparat	ion of modified silica nanoparticles via post-	
		modifica	ation method with APTMS in aqueous media	83
	3.4.7	Preparat	ion of thin film of modified silica nanoparticles	
		via co-co	ondensation method	84
	3.4.8	Preparat	ion of thin film of modified silica via post-modification	
		method	with APTES in aqueous media	84
	3.4.9	Preparat	ion of thin film of silica via sol-gel procedure	84
3.5	Exper	imental pr	ocedures: Part 3	85

3.5.1	Standard procedure for the preparation of polypyrrole	
	(PPy) bulk powder	85
3.5.2	General and standard procedure for the preparation of silica-	
	polypyrrole nanocomposite via a bulk polymerization	85
3.5.3	Standard procedure for the preparation of silica-polypyrrole	
	nanocomposite via a bulk polymerization using different solvents	86
3.5.4	General procedure for the preparation of core-shell system of	
	in-situ modified silica (APTES-SiO ₂)-polypyrrole nanocomposite	
	by initial deposition of pyrrole	86
3.5.5	General procedure for the preparation of core-shell system	
	of in-situ modified silica (APTES-SiO ₂)-polypyrrole	
	nanocomposite by initial deposition of FeCl ₃ .6H ₂ O	87
3.5.6	Standard procedure for the preparation of core-shell system	
	of silica-polypyrrole nanocomposite by initial deposition	
	of pyrrole	88
3.5.7	Standard procedure for the preparation of core-shell system	
	of silica-polypyrrole nanocomposite by initial deposition of	
	FeCl ₃ .6H ₂ O	88
3.5.8	Standard procedure for the preparation of core-shell system	
	of different post-modified silica (APTES-p-SiO ₂)-polypyrrole	
	nanocomposites by initial deposition of pyrrole	88
3.5.9	Standard procedure for the preparation of core-shell system	
	of <i>in-situ</i> modified silica (APTES-SiO ₂)-polypyrrole	
	nanocomposite by initial deposition of pyrrole in different	
	deposition times	89
3.5.10	Standard procedure for the preparation of core-shell system	
	of <i>in-situ</i> modified silica (APTES-SiO ₂)-polypyrrole	
	nanocomposite by initial deposition of FeCl ₃ .6H ₂ O in different	
	deposition times	89
3.5.11	Standard procedure for the preparation of core-shell system	
	of <i>in-situ</i> modified silica (APTES-SiO ₂)-polypyrrole	
	nanocomposite by initial deposition of pyrrole in the	
	presence of $(NH_4)_2S_2O_8$ as an oxidant	89

3.5.12	Standard procedure for the preparation of core-shell system	
	of <i>in-situ</i> modified silica (APTES-SiO ₂)-polypyrrole	
	nanocomposite by initial deposition of $(NH_4)_2S_2O_8$ as an	
	oxidant	90
3.5.13	Standard procedure for the preparation of core-shell system	
	of <i>in-situ</i> modified silica (APTES-SiO ₂)-polypyrrole	
	nanocomposite by initial deposition of pyrrole in the	
	presence of p-toluene sulfonic acid as a dopant	90
3.5.14	Standard procedure for the preparation of core-shell system	
	of <i>in-situ</i> modified silica (APTES-SiO ₂)-polypyrrole	
	nanocomposite by initial deposition of FeCl ₃ .6H ₂ O in the	
	presence of p-toluene sulfonic acid as a dopant	90
3.5.15	Standard procedure for the preparation of core-shell system	
	of <i>in-situ</i> modified silica (APTES-SiO ₂)-polypyrrole	
	nanocomposite by initial deposition of pyrrole in the	
	presence of carbon nanotube (CNT) as a dopant	91
Charac	eterization techniques	91
3.6.1	Transmission Electron Microscopy (TEM)	91
3.6.2	Scanning Electron Microscopy (SEM) with Energy	
	Dispersive X-ray spectroscopy (EDX)	92
3.6.3	Atomic Force Microscopy (AFM)	92
3.6.4	Photoluminescence (PL) spectroscopy	92
3.6.5	Raman spectroscopy	93
3.6.6	Fourier Transform Infrared spectroscopy (FTIR)	93
3.6.7	Solid state reflectance UV-Vis spectroscopy	93
3.6.8	Fourier Transform Nuclear Magnetic Resonance (FT-NMR)	93
3.6.9	X-ray diffraction (XRD)	94
3.6.10	Porosimeter	94
3.6.11	Thermogravimetric Analysis (TGA), TGA-FTIR and TGA-MS	94
3.6.12	Differential Scanning Calorimetry (DSC)	95
3.6.13	CHNOS analyzer	95
3.6.14	Solid conductometer	96
3.6.15	Optical Reflectometer (Filmetrics)	96
3.6.16	Digital Density Meter (Pycnometer)	96

3.6

CHAPTER 4: RESULTS AND DISCUSSION – PREPARATION AND CHARACTERIZATION OF SILICA NANOPARTICLES

4.1	Introd	uction	98
4.2	Prepa	ration and characterization of silica nanoparticles	98
	4.2.1	Effect of experimental parameters on the particle size and	
		morphology	99
	4.2.2	Effect of mixing mode	102
	4.2.3	Effect of drying technique	106
	4.2.4	SEM analysis	107
	4.2.5	Porosimetry	108
	4.2.6	Silanol concentration (δ_{OH}) and silanol number (α_{OH})	113
	4.2.7	Thermal analysis	114

REFERENCES

119

CHAPTER 5: RESULTS AND DISCUSSION – ORGANO-MODIFICATION OF SILICA NANOPARTICLES

5.1	Introd	ucation	121
	5.1.1	Effect of loading of coupling agent	124
	5.1.2	Solvent effect	125
	5.1.3	Gelling time	127
	5.1.4	FT-IR analysis	128
	5.1.5	Elemental analysis	129
	5.1.6	Solid State ²⁹ Si and ¹³ C CP/MAS NMR analysis	131
	5.1.7	Thermal analysis	133
5.2	Optica	Il properties of organo-modified silica nanoparticles produced via	
	co-cor	ndensation method	134
	5.2.1	XRD analysis	134

	5.2.2	Photoluminescence analysis	135
	5.2.3	Solid state ultraviolet (UV) analysis	139
	5.2.4	Raman spectroscopy analysis	141
5.3	A con	nparison study on <i>in-situ</i> modification and post-modification of	
	silica	nanoparticles	143
	5.3.1	Characterization	144
		5.3.1.1 FT-IR analysis	144
		5.3.1.2 The mass loading percentage onto silica particles	145
		5.3.1.3 TGA and DTG analysis	148
	5.3.2	Physical properties	152
		5.3.2.1 Porosity	152
		5.3.2.2 Apparent density	157
	5.3.3	Morphological properties (TEM and SEM analysis)	157
	5.3.4	Optical properties	161
		5.3.4.1 Photoluminescence analysis	161
		5.3.4.2 Solid state ultraviolet (UV) analysis	163
	5.3.5	Surface topography	164
REFE	RENCE	ES	167

CHAPTER 6: RESULTS AND DISCUSSION – PREPARATION AND CHARACTERIZATION OF SiO₂@POLYPYRROLE NANOCOMPOSITES

6.1	Introd	ucation	171
6.2	Optim	nization	172
	6.2.1	Preparation methods	172
		6.2.1.1 The effect of preparation method (method A)	
		on particle size, conductivity and morphology of conducting	
		polymer-based nanocomposites	173
		6.2.1.2 The effect of preparation method (method B and C)	
		on particle size, conductivity and morphology of conducting	
		polymer-based nanocomposites	183

	6.2.1.3 Type of deposited species	187
	6.2.1.4 Deposition time of reactants	189
	6.2.1.5 Type of oxidant	189
	6.2.1.6 Dopant effect	190
	6.2.1.7 Comparison between different modified silica prepared	
	via in-situ and post modification	191
6.2.2	Characterization	194
	6.2.2.1 FT-IR analysis	194
	6.2.2.2 Solid state NMR	195
	6.2.2.3 Ultraviolet analysis	198
	6.2.2.4 Thermal behavior by TGA-FTIR results	199
	6.2.2.5 Photoluminescence (PL) analysis	201

REFERENCES

202

CHAPTER 7: GENERAL CONCLUSION AND RECOMMENDATION FOR FUTURE WORK

7.1	General Conclusion	205
7.2	Recommendation for Future Research	209

APPENDIX

LIST OF PUBLICATIONS

LIST OF TABLES

Table 3.1	Optimized experimental parameters for the preparation of silica nanoparticles	Page 78
Table 3.2	Optimized experimental parameters for the preparation of modified-silica nanoparticles	81
Table 3.3	The different mixtures of precursor for the preparation of modified silica via co-condensation method	82
Table 4.1	Effect of feed rate of ammonia on particle size of silica obtained via freeze drying	100
Table 4.2	Effect of mixing mode on particle size of silica obtained via freeze drying	103
Table 4.3	Effect of drying process on particle size of silica obtained via freeze drying	107
Table 4.4	The effect of drying process on BET surface area, pore volume, and pore diameter of the samples	109
Table 4.5	The effect of drying process on number of hydroxyl content	118
Table 5.1	Optimized experimental parameters for the preparation of modified silica nanoparticles	121
Table 5.2	The effect of APTES content on particle size of modified silica particle	124
Table 5.3	The effect of different solvents on particle size distribution of modified silica particle	127

- Table 5.4Elemental composition of modified silica that prepared by146different methods
- Table 5.5The mass loss of modified silica at different temperature150range
- Table 5.6The BET surface area, pore volume, pore diameter, density152and particle size of the modified silica
- Table 5.7Some parameters of silica, *in-situ* modified silica and post-166modified silica thin film
- Table 6.1The conductivity of different PPy-based silica and modified174silica nanocomposites prepared under different procedures
- Table 6.2The conductivity of different silica and modified silica-PPy184core-shell nanocomposites prepared under different procedure
- Table 6.3The elemental composition of different silica and modified188silica-PPy core-shell nanocomposites that prepared under
different procedures
- Table 6.4The mass losses of different modified silica, before and after192polymerization with PPy

LIST OF FIGURES

		Page
Figure 1.1	Particle growths under the (A) acidic and (B) basic sol-gel conditions.	4
Figure 1.2	The structure of some conducting polymers	12
Figure 1.3	The Oxidative polymerization of PPy	14
Figure 2.1	Typical polymerization mechanism of pyrrole adopted from Jang (2006)	54
Figure 4.1	FT-IR spectra of silica particles obtained from freeze drying (a) before calcinations (FD) and (b) after calcination (FD-C)	99
Figure 4.2	Particle size distribution (PSD) of silica nanoparticles obtained via freeze drying in different ammonia feed rate	101
Figure 4.3	TEM images of silica nanoparticles obtained via freeze drying in different ammonia feed, (A) 0.01, (B) 0.02 (C) and 0.03 mL min ⁻¹	102
Figure 4.4	Flow chart for nanosilica preparation by different mixing modes	103
Figure 4.5	Particle size distribution (PSD) of silica nanoparticles obtained from freeze drying in different mixing modes	104
Figure 4.6	TEM images of silica nanoparticles obtained from freeze drying in different mixing modes, (A) Mode-A, (B) Mode-B and (C) Mode-C	104

- Figure 4.7 Particle size distribution (PSD) of silica nanoparticles in 107 different drying process: freeze drying (FD-C) and heat drying (HD-C)
- Figure 4.8 SEM images of silica nanoparticles obtained via (A) heat 108 drying (HD-C) and (B) freeze drying (FD-C)
- Figure 4.9 The effect of drying process on volume of adsorbate (based on 110 BET model) onto the surface area of silica samples
- Figure 4.10 Nitrogen sorption isotherms of silica nanoparticles obtained 111 via freeze (FD-C) and heat (HD-C) drying
- Figure 4.11 (A) Pore area and (B) pore volume distribution (BJH 112 method) of silica nanoparticles obtained via freeze (FD-C) and heat (HD-C) drying
- Figure 4.12 TG and DTG thermograms of silica particles obtained (A) via 116 heat drying, before calcination (HD) and after calcination (HD-C) (B) via freeze drying, before calcination (FD) and after calcination (FD-C)
- Figure 4.13 DSC thermograms of silica particles obtained via heat drying 117 (HD) and freeze drying (FD)
- Figure 5.1 Flow chart for modified nanosilica preparation via a cocondensation method
- Figure 5.2 (A) Particle size distribution modified silica nanoparticles 123 prepared in ethanol and (B) TEM image of modified silica
- Figure 5.3 Particle size distribution (PSD) of modified silica 125 nanoparticles with different amount of APTES

XV

- Figure 5.4 (A) Particle size distribution (PSD) of modified silica 126 nanoparticles that were synthesized with different solvent. (B)
 TEM image of modified silica nanoparticles prepared in methanol
- Figure 5.5 The effect of gelling time on the addition of APTES 128
- Figure 5.6 FTIR spectra of precursors (a) TEOS and (b) APTES, and (c) 129 modified silica particles
- Figure 5.7 (A) EDX spectrum of modified silica nanopartciles and (B) 130 SEM image of modified silica
- Figure 5.8 ²⁹Si CP/MAS solid state NMR spectra of modified silica 132 particles
- Figure 5.9 ¹³C CP/MAS solid state NMR spectrum of modified silica 133 particles
- Figure 5.10 TG and DTG thermograms of modified silica nanoparticles 134 via co-condensation method
- Figure 5.11 XRD pattern of silica and modified silica nanoparticles 135
- Figure 5.12 Room temperature PL spectra of pure silica (dash-line) and 138 modified silica (solid-line) nanoparticles
- Figure 5.13 Optical absorption of organo-modified silica and pure silica in 140 (A) UV range and (B) near vacuum UV region
- Figure 5.14 Raman spectra of pure silica (dash-line) and modified silica 142 (solid-line) nanoparticles

- Figure 5.15 Flow chart for the preparation of modified silica that obtained via post-modification methods in (A) non-aqueous and (B) 144 aqueous solvents
- Figure 5.16 FTIR spectra of modified silica particles (a) SiO₂-APTES, (b) APTMS-p-SiO₂-E, (c) APTES-p-SiO₂-E, (d) APTES-p-SiO₂- 145 T
- Figure 5.17 The (A) TG and (B) DTG of different modified silica (a) APTMS-p-SiO₂-E, (b) APTES-p-SiO₂-E, (c) APTES-p-SiO₂- 149 T, (d) SiO₂-APTES prepared via different methods
- Figure 5.18 Nitrogen sorption isotherms of (A) post-modified and (B) *insitu* modified silica nanoparticles obtained via different 155 modification processes
- Figure 5.19 Pore volume distributions (BJH method) of (A) post-modified and (B) *in-situ* modified silica nanoparticles obtained via 156 different modification processes
- Figure 5.20 TEM images of different modified silica nanoparticles obtained via *in-situ* modification and post-modification 159 methods. (A) APTES-p-SiO₂-T, (B) APTES-p-SiO₂-E, (C) APTMS-p-SiO₂-E, (D) SiO₂-APTES
- Figure 5.21 TEM images of (A) APTES-p-SiO₂-T, (B) APTMS-p-SiO₂-E obtained via post-modification methods in non-aqueous and 160 aqueous media
- Figure 5.22 SEM images of different modified silica nanoparticles that obtained via *in-situ* modification and post-modification 161 methods. (A) SiO₂-APTES, (B) APTES-p-SiO₂-T, (C) APTES-p-SiO₂-E, (D) APTMS-p-SiO₂-E.

- Figure 5.23 The PL spectrum of different modified silica particle, (a) APTES-p-SiO₂-T, (b) APTES-p-SiO₂-E, (c) APTMS-p-SiO₂- 162 E
- Figure 5.24 The UV spectra of different modified silica (a) APTES-p-SiO₂-T, (b) APTES-p-SiO₂-E, (c) APTMS-p-SiO₂-E at 164 different wavelength range 200-650 nm
- Figure 5.253D and 2D AFM images of thin film of (A) pure silica, (B) in-
situ modified silica and (C) post-modified silica165
- Figure 6.1 Flow chart for the preparation of SiO₂-PPy composites (mode A) and SiO₂-APTES-PPy (Mode B and C) core-shell 173 nanocomposites via a dispersion oxidative polymerization
- Figure 6.2 TEM images of PPy-based silica nanocomposites: (A) SiO₂PPy (Ref), (B) SiO₂-PPy (Ultrasonic), (C) SiO₂-PPy (Mixing), 177
 (D) SiO₂-PPy-Et, (E) SiO₂-PPy-Met
- Figure 6.3 TEM images of PPy-based modified silica nanocomposites:
 (A) SiO₂-APTES-PPy, (B) SiO₂-APTES-PPy (EtOH), (C) 179
 SiO₂-APTES-PPy (MeOH), (D) APTES-p-SiO₂-T-PPy, (E)
 APTES-p-SiO₂-E-PPy, (F) APTMS-p-SiO₂-E-PPy
- Figure 6.4 SEM images of (A) PPy, (B) SiO₂-PPy and (C) SiO₂-APTES-PPy nanocomposites 181
- Figure 6.5 EDX of SiO₂-PPy nanocomposites in (A) spot on the particle and (B) area around the particle 182
- Figure 6.6 TEM images of (A) SiO₂@PPy and (B) SiO₂-APTES@PPy core-shell nanocomposites at different magnifications 3000 185 and (C) 60000

Figure 6.7	SEM images of (A) $SiO_2@PPy$, (B) SiO_2 -APTES@PPy and	
	(C) EDX of SiO ₂ -APTES@PPy core-shell nanocomposites	186
Figure 6.8	TGA thermograms of nanocomposite particles, (a) APTMS-p-SiO ₂ -Et@PPy, (b) APTES-p-SiO ₂ -E@PPy, (c) APTES-p-SiO ₂ -T@PPy, (d) SiO ₂ -APTES@PPy, (e) PPy	193
Figure 6.9	FTIR spectra of (a) $SiO_2@PPy$, (b) PPy, (c) APTES-p-SiO ₂ -T@PPy, and (d) SiO_2 -APTES@PPy	195
Figure 6.10	²⁹ Si CP/MAS solid state NMR spectra of silica- and modified silica-PPy nanocomposite	196
Figure 6.11	¹³ C CP/MAS solid state NMR spectra of SiO ₂ -APTES-PPy nanocomposites and SiO ₂ -APTES nanoparticles	197
Figure 6.12	The UV spectra of (a) APTES-p-SiO ₂ -T@PPy, (b) PPy, (c) SiO ₂ -APTES@PPy and (d) SiO ₂ @PPy	199
Figure 6.13	TG/DTG thermograms of SiO ₂ -APTES@PPy nanocomposite particles	200
E'		202

Figure 6.13 Room temperature PL spectra of (a) calcined SiO₂- 202 APTES@PPy and (b) SiO₂-APTES@PPy nanocomposite particles

LIST OF SYMBOLS, ABBREVIATIONS AND NOMENCLATURE

Symbols	Descriptions
AFM	Atomic Force Microscopy
$\alpha_{\rm OH}$	Silanol number
APTES	3-aminopropyltriethoxysilane
APTMS	3-aminopropyltrimethoxysilane
BET	Brunauer-Emmet-Teller
BJH	Barett-Joyner-Halenda
CNT	Carbon nanotube
CP/MAS NMR	Cross-polarization magic angle spinning NMR
Da	Apparent density
ΔH	Enthalpy
δ_{OH}	Silanol concentration
DSC	Differential Scanning Calorimetry
EDX	Energy dispersive X-ray
eV	Electron volt
FD	Freeze drying
FTIR	Fourier transform infrared
HD	Heat drying
λ_{max}	Lambda maximum
LO	Longitudinal optical
N _A	Avogadro number (6.022×10^{23})
NBOHC	Non-bridging oxygen hole center
NMR	Nuclear magnetic resonance
ν	Stretching vibration
ODC	Oxygen deficient centers
Oxi	Oxidant
PAni	Polyanilline
PL	Photoluminescence
PSD	Particle size distributions
Py	Pyrrole

Symbols	Descriptions
РРу	Polypyrrole
R	[H ₂ O]/[TEOS] ratio
R′	[H ₂ O]/[TEOS+APTES] ratio
R _a	Roughness
SEM	Scanning electron microscopy
σ	Standard deviation of particle size
STE	Self-trapped exciton
Т	Temperature (°C)
TEM	Transmission electron microscopy
TEOS	Tetraethylorthosilicate
TGA	Thermogravimetric analysis
TMS	Tetramethylsilane
ТО	Transverse-optical
Ts	<i>p</i> -toluenesulfonic acid
UV-Vis	Ultraviolet-visible
XRD	X-ray diffraction

SINTESIS, PENGUBAHSUAIAN NANOZARAH SILIKA SERTA PENGGUNAANNYA DALAM PEMFABRIKAN SILIKA NANOKOMPOSIT BERASASKAN POLIPIROL

ABSTRAK

Kajian ini menerangkan tentang pengubahsulan penyediaan silika zarah nano secara sol-gel. Keupayaan untuk mengawal saiz dan taburan zarah didapati bergantung pada mod campuran reaktan dan teknik pengeringan. Campuran tetraetoksilana (TEOS) dan etanol, diikuti dengan penambahan air (Mod - A) menghasilkan serbuk mono serakan dengan purata saiz zarah 10.6 ± 1.40 nm dan taburan saiz yang sempit. Selanjutnya, teknik pengeringan kering beku (FD) meningkatkan lagi kualiti serbuk tersebut berbanding dengan penggunaan teknik pengeringan haba (HD). Serbuk ini terdiri daripada struktur teratur silika nano zarah dengan tahap kecutan yang rendah, saiz dan isipadu liang yang lebih besar serta entalpi endoterma yang besar (ΔH_{FD} = - 688 J g⁻¹ vs. $\Delta H_{HD} = -617 \text{ J g}^{-1}$, yang memudahkan lagi pelepasan keluar air terjerap. Sehubungan ini, suatu kaedah mudah dan pantas dalam menyediakan silika organo terubah suai dalam julat saiz nano dengan kumpulan amina hujung secara kondensasi bersama telah dilakukan. Nano silika terubah suai organo terdiri daripada serbuk amorfus dengan saiz zarah ~ 60 nm, mono sebaran yang baik dan pengagregatan yang rendah. Kecacatan struktur bagi nano zarah silika organo terubah suai telah dikaji melalui penyerapan optik dan fotopendarcahaya dalam julat ultra ungu vakum (VUV), UV boleh nampak dan hampir IR. Analisis Raman juga menunjukkan ciri-ciri optik berkaitan kecacatan permukaan. Nano zarah silika tersedia (bersaiz ~ 10 nm) telah ditukar secara pasca pengubahsuaian kepada saiz baru 11-13 nm, tetapi dengan pengagregatan lebih tinggi dan kehadiran organosilana swakondensi. Kandungan spesis organik dengan kefungsian amino dicerap bagi silika terubah suai, menghasilkan

kehidrofobiaan permukaan yang lebih banyak yang dapat meningkatkan lagi keserasian dan serakan silika organo terubah suai menjadi matriks polimer dalam pemfabrikatan nano komposit. Akhir sekali, sistem teras kelompang nano komposit silika-polipirola dengan saiz zarah 60-70 nm telah disintesiskan melalui pempolimeran penyebaran oksidatif menggunakan silika tulen dan silika terubah suai yang disediakan sebelum ini. Zarah-zarah tersebut menunjukkan tahap sebaran lebih tinggi dan pengagregatan lebih rendah tanpa menggunakan sebarang penstabil cas dan sterik. Prapengubahsuaian permukaan silika menunjukkan kesan ketara terhadap keberkesanannya sementara jumlah pembebanan jisim polipirola pada permukaan silika terubah suai mengakibatkan kekonduksian elektrik yang tinggi berbanding silika tulen. Kaedah penyediaan berdasarkan mod pra-endapan pirola atau FeCl₃.6H₂O sebagai zat pengoksidaan merupakan satu lagi faktor penting terhadap keberkesanan pembebanan jisim polipirola pada permukaan silika terubah suai. FeCl₃.6H₂O ialah agen pengoksidaan yang lebih berkesan bagi pempolimeran pirola berbanding dengan penggunaan satu lagi agen biasa iaitu $(NH_4)_2S_2O_8$. Penambahan p-toluena sulfonat sebagai bahan dop anion semasa proses pempolimeran telah meningkatkan lagi kekonduksian polipirola.

SILICA NANOPARTICLES, SYNTHESIS, MODIFICATION AND THEIR APPLICATIONS IN THE FABRICATION OF POLYPYRROLE-BASED SILICA NANOCOMPOSITES

ABSTRACT

A modified preparation of silica nanoparticles via sol-gel process is described. The ability to control the particle size and distribution is found to be highly dependent on mixing modes of the reactants and drying techniques. The mixture of tetraethoxysilane (TEOS) and ethanol followed by addition of water (Mode-A) produce monodispersed powder with an average particle size of 10.6 ± 1.40 nm and a narrow size distribution. The freeze drying technique (FD) further improves the quality of the powder compared to heat drying (HD). The powder consists of a well-order structure of silica nanoparticles with low shrinkage, bigger pore size and volume and large endothermic enthalpies ($\Delta H_{FD} = -688 \text{ J g}^{-1}$ vs. $\Delta H_{HD} = -617 \text{ J g}^{-1}$), leading to easy escape of physically adsorbed water. Then, an easy and swift pathway in the preparation of organo-modified silica in nanosize range with amine terminal group via cocondensation method has been obtained. The organo-modified nanosilica consists of amorphous powder with particle size of ~ 60 nm, highly monodispersed and low aggregation. Structural defects in organo-modified silica nanoparticle have been studied via optical absorption and photoluminescence in the range of vacuum ultraviolet (VUV), UV visible, and near-IR. Raman analysis also demonstrates the surface defectrelated optical properties. Post-modification of as-prepared silica nanoparticles (~ 10 nm in size) are converted into organo-functionalized silica with new size of 11-13 nm, but with higher aggregation and existence of self-condensed of organosilanes. Higher organic species content with amino functionality is observed for *in-situ* modified silica, leading to more surface hydrophobicity that can improve the compatibility and

dispersion of organo-modified silica into polymeric matrix in nanocomposite fabrication. Finally, the core-shell system of silica-polypyrrole nanocomposite with particle size of 60-70 nm has been synthesized via oxidative dispersion polymerization using pure silica and modified silica which was previously prepared. The particles demonstrate higher dispersion and lower aggregation without employing any charge and steric stabilizers. Pre-surface modification of silica shows significant effect on efficiency of polypyrrole deposition while the amount of mass loading of polypyrrole on the surface of modified silica compared to that in pure silica leading to higher electrical conductivity. The preparation method based on the mode of pre-deposition of pyrrole or FeCl₃.6H₂O as oxidant is another important factor on efficiency of mass loading of polypyrrole on the surface of modified silica. Addition of *p*-toluene sulfonate as anion dopant during polymerization process has highly enhanced the conductivity of polypyrrole.

CHAPTER 1: INTRODUCTION

1.1 Nanoparticles

Nanostructures or nanoscale materials are any solid materials that have a nanometer dimension. Nanoparticles are generally considered to be a number of atoms or molecules bonded together with a radius of < 100 nm. These nanosize solid particles could be nanocystalline, an aggregate of crystallites, or a single crystallite. The physical, chemical, and electronic properties of nanoparticles depend strongly on the number and type of atoms that make up the particle. It is found that color, reactivity, stability, and magnetic behavior all depend on the particle size. In some instances, an entirely new behavior, which is not seen in the bulk, has been observed such as magnetism in clusters that are constituted from nonmagnetic atoms (Wang, 2000; Poole & Owens, 2003; Köhler & Fritzsche, 2004; Rao *et al.* 2004; Sarid *et al.* 2004; Schmid, 2004; Rao & Cheetham, 2006). There are some preparation methods for nanoparticles such as laser evaporation, radio frequency (RF) plasma, thermolysis and chemical methods (Sarid *et al.*, 2004; Sharma *et al.*, 2004; Skandan & Singhal, 2006).

1.2 Silica nanoparticles

Silica is an inorganic polymer [(SiO₂)n] formed via condensation of silicic acid as a fully hydrolyzed monomer [Si(OH)₄]. Silica is generally used as a short convenient designation for "silicon dioxide" in all its crystalline, amorphous, and hydrated or hydroxylated forms (Iller, 1979). Silica has two different functional groups on its surface; silanol (Si—OH) and siloxane (Si—O—Si) groups. These two groups

substantially influence the surface properties and subsequently the application properties. The fully hydroxylated surface of silica generally has ~ 4.6 silanol groups per nm² which results in the hydrophilic character of products (Brinkmann *et al.*, 2006). There are various methods for the preparation of nanosilica, among them, the well-known / used is sol-gel technique.

1.2.1 Sol-gel technique

The synthesis of silica by sol-gel technique involves the evolution of inorganic network (Si-O-Si) through the formation of a colloidal suspension (sol) and gelation of the sol, forming a network in a continuous liquid phase (gel) (Brinker & Scherer, 1990). At the functional group level, three reactions are generally used to describe the sol-gel process: hydrolysis, alcohol condensation, and water condensation. This general reaction scheme can be seen as follows;

Hydrolysis

$$\equiv \text{Si-O-R} + \text{H}_2\text{O} \rightarrow \equiv \text{Si-OH} + \text{R-OH}$$
(1)

Water condensation

$$\equiv \text{Si-OH} + \text{HO-Si} \implies \equiv \text{Si-O-Si} \implies + \text{H}_2\text{O}$$
(2)

Alcohol condensation

 $\equiv \text{Si-O-R} + \text{HO-Si} \implies \equiv \text{Si-O-Si} \implies + \text{R-OH}$ (3)

Under most conditions, condensation commences before hydrolysis is completed. However, parameters such as pH, $[H_2O]/[TEOS]$ molar ratio (R), and catalyst can force the completion of hydrolysis before the condensation begins. Additionally, since water and alkoxide are immiscible, a mutual solvent such as an alcohol is utilized. Thus, in the presence of this homogenizing agent i.e. alcohol, hydrolysis is facilitated due to the miscibility of the alkoxide and water (Brinker, 1988, 2006). As the number of siloxane bonds increases in the condensation process, the individual molecules are bridged and jointly aggregated in the sol. When the sol particles aggregate, or are inter-knited into a network, a gel is formed. The gel is dried by means of low temperature treatments (25-100 °C) or under a supercritical condition and is converted into dense ceramics (xerogel) or extremely low-density material called "aerogel", respectively (Hench & West, 1990). Upon drying, trapped volatiles (water, alcohol, etc.) are driven off and the network shrinks. It should be emphasized that the addition of solvents and certain reaction conditions as a serious drawback may promote esterification and depolymerization reactions according to the reverse of equations (1), (2), and (3) (Celzard & Marêché, 2002; Coltrain & Kelts, 2006).

In morphological aspect point of view, sol-gel polymerization occurs in three stages: (a) Polymerization of monomers to form particles, (b) Growth of particles, and (c) Linking of particles into chains, and subsequently networks that extend throughout the liquid medium, thickening into a gel. Within the context of these stages, many factors affect the resulting silica network, such as, pH, temperature and time of reaction, reagent concentration, catalyst nature and concentration, [H₂O]/[TEOS] molar ratio (R), aging temperature and time, and drying techniques (Iller, 1979). Thus, by controlling these factors, it is possible to control nucleation and the growth of particle as well as to avoid any undesirable phenomenon such as aggregation and agglomeration.

However, the sol-gel process under acid-catalyzed conditions yields primarily linear or randomly branched polymers which entangles and forms additional branches in gel. On the other hand, the reaction under base-catalyzed conditions yields more highly branched clusters which does not interpenetrate prior to gelation and thus behave as discrete clusters (Brinker & Scherer, 1990) (Figure 1.1).



Figure 1.1 Particle growths under the (A) acidic and (B) basic sol-gel conditions. (Accessed on 14 May 2009, <u>http://www.psrc.usm.edu/mauritz/solgel.html</u>)

1.2.2 Applications of nanosilica

Silica powders can be represented by different functions in different applications, for example, reinforcing, stiffening and hardening of organic polymers, reducing adhesive between solid surfaces, increasing adhesion of adhesives, increasing viscosity in liquid, creating various optical effect, surfactant effect, absorbent, catalyst base, chromatographic column packing and biological application such as dental filling (Hench, 1998; Bergna, 2006; Falcone, 2006; Wang *et al.*, 2008; Yeoh *et al.*, 2009).

1.3 Modification process

In silica, where the siloxane groups are generally chemically inert, the reactivity of silanol groups allows some chemical surface modifications. Thus, the reaction with organosilanes can lead to hydrophobic silica. The attachment of silane coupling agent molecules to colloidal silica particles can prevent the aggregation of the particles due to the steric repulsion of grafted organic groups (Tertykh, 2004). The combination of the properties of organic and inorganic components lead to the preparation of hybrid materials that possess the enormous functional variation of organic chemistry with the advantages of a thermally stable and robust inorganic substrate. On the other hand, the adsorption phenomenon depends on the interaction between the surface of the adsorbent and the adsorbed species. The interaction may be due to, (1) chemical bonding, (2) hydrogen bonding, (3) hydrophobic bonding, (4) van der Waals force (Parida *et al.*, 2006).

Three pathways are available for synthesizing the modified hybrid materials based on organosilica units: (1) the subsequent modification of the surface of a purely inorganic silica material ("grafting"); (2) the simultaneous condensation of corresponding silica and organosilica precursors ("co-condensation") and (3) the incorporation of organic groups as bridging components directly and specifically into the pore walls by the use of bissilylated single-source organosilica precursors ("production of periodic mesoporous organosilica") (Hoffmann *et al.*, 2006).

1.3.1 Post-modification (grafting)

Grafting refers to the subsequent modification of the inner surface of mesostructured silica with organic groups. This process is primarily carried out by the reaction of

organosilanes with the free silanol groups of the surfaces. This method of modification has the advantage that the mesostructure of the starting silica is usually retained, whereas the lining of the walls shows a reduction in the porosity of the hybrid material. If the organosilanes react preferentially at the open pore during the initial stages of the synthetic process, the diffusion of further molecules into the center of the pores can be impaired, which can in turn lead to a nonhomogeneous distribution of the organic groups within the pores and a lower degree of occupation. In extreme cases (e.g., with very bulky grafting species), this can lead to a complete closure of the pores (pore blocking) (Vansant *et al.*, 1995; Hoffmann *et al.*, 2006).

1.3.2 *In-situ* modification (co-condensation method)

An alternative method to one-pot synthesize organic-functionalized mesoporous silica is the co-condensation method (Direct method). It is possible to directly prepare mesostructured silica by the co-condensation of tetraalkoxysilanes with terminal trialkoxyorganosilanes, leading to materials with organic residues anchored covalently to the pore walls. Since the organic functionalities are direct components of the silica matrix, pore blocking is not a problem in the co-condensation method. Furthermore, the organic units can generally and more homogeneously be distributed than in materials synthesized with the grafting process owing to the homogeneous distribution of different organic functionalities in the framework. However, the co-condensation method also has a number of disadvantages as well: in general, the degree of order of the products decreases with the increasing concentration of (R'O)₃SiR in the reaction mixture, which ultimately leads to totally disordered products. An increasing proportion of (R'O)₃SiR in the reaction mixture favors homo-condensation reactions—at the cost of cross-linking co-condensation reactions

with the silica precursors (Hoffmann *et al.*, 2006). Therefore, by controlling the amount of organosilanes, it could constrain any undesirable formation of homo-condensed product.

1.3.3 Incorporation modification

The synthesis of organic-inorganic hybrid materials by hydrolysis and condensation reactions of bridged organosilica precursors of the type (R'O)₃Si-R-Si(OR')₃ is known via sol-gel chemistry. The organic units in this case incorporate in the three-dimensional network structure of the silica matrix through two covalent bonds and thus are distributed totally homogeneously in the pore walls. These materials, which are obtained as porous aerogel and xerogels, can have large inner surface areas as well as high thermal stability but generally exhibit completely disordered pore systems with a relatively wide distribution of pore radii. This method is employed for the preparation of a new class of mesostructured organic-inorganic hybrid materials-periodic mesoporous organosilicas (PMOs) (Asefa *et al.*, 1999; Inagaki *et.al.*, 1999; Melde *et al.*, 1999). Recently, microwave heating was employed for the preparation of periodic mesoporous organosilica material with smaller and uniform particle size (Yoon *et al.*, 2008).

1.4 Nanocomposites

Many efforts have focused on the integration of a variety of nanoscopic materials into polymers and the impact of these composites on a broad spectrum of applications, from engineering plastics to electronic materials to biotechnology. Significant interest in the use of well-ordered nanoparticles as fillers within polymer materials are expected to provide access to new materials (known as organicinorganic nanocomposite) that combine the unique physical properties of the inorganic particles (rigidity, high surface hardness, high modules, strength, high thermal stability, UV-shielding property, and high refractive index, etc.) with the superior processibility of the polymers (flexibility, ductility, transparency, mouldability, low density, dielectric strength, etc.) (Ajayan *et al.*, 2003; Sill *et al.*, 2004; Koo, 2006).

Organic-inorganic nanocomposites, which are based on organic polymer and inorganic mineral exhibit the enhanced performance properties compared to conventional composites owing to the maximized interfacial contact between the organic and inorganic phases in comparison with bulk materials (Seo & Kim, 2005). However, it is very difficult to manufacture nanocomposites with the nanoparticles dispersed uniformly as nanoparticles have very high surface energies and are prone to agglomerating into larger particles during the preparation of the materials (Xu *et al.*, 2003). The most important approach that can be employed to solve the problems is a surface organo modification of nanoparticles in order to promote compatibility. Therefore, the properties of a composite material depend not only upon the properties of the individual component phases (matrix, filler, interphase), but also upon their interaction (Kashiwagi *et al.*, 2003). Owing to the poor compatibility between the organic matrix and inorganic fillers, some efforts are necessary in order to enhance the compatibility between the fillers and the hydrophobic polymer matrix during the preparation of nanocomposites (Y. -L. Liu *et al.*, 2003).

Generally, there are two typical types of interactions in organic-inorganic nanocomposites, depending on the strength or level of interaction between organic and inorganic phases: one involving physical or weak phase interaction (e.g., hydrogen bonding, van der Waals forces) and another possessing a strong chemical covalent or ionic-covalent bond between the organic and inorganic phases (Chen *et al.*, 2005). These forces can strongly affect on physical, mechanical and thermal properties of nanocomposites.

1.4.1 Preparation methods of nanocomposites

To provide well dispersion of these organic-modified nanoparticles in a polymer matrix, there must be favorable interactions between the nanoparticle and the polymer. Several approaches have been investigated for this aim, such as (i) sol-gel process, (ii) *in-situ* polymerization, and (iii) direct solution blending (Yang & Nelson, 2004).

1.4.1.1 Sol-gel process for nanocomposites preparation

In sol-gel process, nanoscale particles of the inorganic material can be formed *in-situ* and the interaction between the inorganic compound and the polymer can be controlled in order to avoid the formation of voids between both materials. The inorganic moiety can be inserted into the polymer backbone by a reaction between functional groups of the polymer and the alkoxysilane monomers, or, the inorganic compound can be formed *in-situ* without covalent crosslinking with the polymer backbone (Gomes *et al.* 2005).

1.4.1.2 In-situ polymerization process for nanocomposites preparation

In *in-situ* polymerization, nanosilica particle is first dispersed into polymer precursors (monomers), and subsequently polymerization process is conducted.

Dispersing *in-situ* polymerization may be the most desirable method for preparing nanocomposites as the types of nanoparticles and the nature of polymer precursors can vary in a wide range, depending on the requirement (Xu *et al.* 2003).

1.4.1.3 Blending process for nanocomposites preparation

Although nanocomposites can be prepared by simply blending the nanoparticles with base polymers via high shear stirring or ball milling, the dispersing degree of nanoparticles and the interfacial adhesion were obviously insufficient in order to obtain desirable material properties (Qi *et al.* 2006).

Among these three approaches, the *in-situ* polymerization method has the advantages of ease of handling, i.e. a relatively quick process, and usually offers better performance for the final products (Yang & Nelson, 2004).

1.4.2 Core-shell nanocomposites

Core-shell or nano-shell particles constitute a special class of nanocomposite materials. They can also be classified as hybrid nanostructures. In this system, the particles of one material are coated with a thin layer of another material using specialized procedures. They can show modified and improved properties than their single component counterparts or nanoparticles of the same size. Their properties can be modified into either constituting materials or core-to-shell ratio. The term of nanoshell is used especially when the thickness of the shell is ca. 1-20 nm. They are frequently referred as core-shell or core@shell particles (Caruso, 2001; Zhang *et al.*, 2004).

1.4.2.1 Preparation methods for core-shell nanocomposites

Preparation of nanoshell particles involves multistep synthesis procedure. It requires a highly controlled and sensitized synthesis protocol to ensure complete coverage of core particles with the shell materials. There are various methods to fabricate coreshell structure, e.g. precipitation, grafted polymerization, microemulsion, reverse micelle, sol-gel condensation, and layer-by-layer adsorption techniques. Although several methods have been established, it is still difficult to control the thickness and homogeneity of the coating. If the reaction is not properly controlled, it eventually leads to aggregation of the core particles, formation of separate particles of shell materials or incomplete coverage (Aguiar *et al.*, 1999; Li & Stöver, 2000; See *et al.*, 2005; Gong *et al.*, 2009).

1.4.2.2 Application of core-shell nanocomposites

The coating of colloidal particles with shell offers the most simple and versatile way of modifying their surface, chemical, optical, magnetic, and catalytic properties. A variety of applications such as fluorescent diagnostic labels, catalysis, avoiding photo degradation, photoluminescence enhancement, photonic crystals creation, bio-conjugates preparation, chemical and colloidal stability have been found (Schärtl, 2000; Caruso, 2001; Zhong & Maye, 2001; Haag, 2004; Basinska, 2005).

1.5 Conducting polymers

Conductive polymers or organic metals are good conductors of electricity. Polymers with conjugated π -electron backbone can be doped to produce materials that exhibit electrical conductivities approaching those of metals. They have the electronic properties of a semiconductor as well as the mechanical properties of polymers. In

addition, conducting polymers offer advantages over metals and other semiconducting materials owing to their good mechanical properties, stability, ease of processing, and relatively low price (Carswell & Grady, 2004). Polyacetylene (PA), poly(*p*-phenylene) (PPP), poly(*p*-phenylene sulfide) (PPS), polypyrrole (PPy), polythiophene and polyaniline (PAni) have been identified as intrinsically conducting polymers. A number of different charge-transfer agents, known as dopants have been used to improve the conductivity of such polymers, such as AsF₅, I_2 , and BF_3 . The doping of conjugated polymers generates high conductivity by increasing the carrier concentration via oxidation or reduction with electron acceptor or donor, respectively (Vernitskaya & Efimov, 1997; Wallace et al., 2003; Filocamo & Grinstaff, 2004). Figure 1.2 represents the structure of some common conducting polymers.



PPP



PPv

Polythiophene

PAni

NH n

PPS

Figure 1.2 The structure of some conducting polymers

1.5.1 Polypyrrole

Polypyrrole (PPy) a promising conducting polymer has widely found of particular interest, owing to their high polarizability, superior conductivity, stability in the oxidized state and interesting redox properties (Vernitskaya & Efimov, 1997). Polypyyrole is an inherently conducting polymer which contains one positive charge delocalized over three or four pyrrole residues in the polymer backbone. These positive charges are counter-balanced by anions (e.g. Cl⁻, NO_3^- , $SO_4^{2^-}$...), called "dopants," that are incorporated during synthesis. PPy has stiff and inflexible polymer chains and can be considered to be a lightly cross-linked (Wallace *et al.*, 2003). PPy is generally amorphous structure that depends on the degree of polymerization where the color changes from yellow to the blue and ultimately, black. Apart from this, the structural ordering in polymeric layers can strongly affects on the conductivities (Vernitskaya & Efimov, 1997; Cho *et al.*, 2007).

1.5.2 Preparation methods of Polypyrrole

The simplicity of the synthetic procedures and availability of the initial monomers are attractive features of PPy. PPy as a high conductive and environmentally stable polymer, has been extensively synthesized via chemical and electrochemical (Electropolymerization) approach (Vernitskaya & Efimov, 1997; Cho *et al.*, 2007).

In chemical method, polypyrrole prepared by oxidation of the monomer with chemical oxidants has a form of black powder. Aqueous or anhydrous FeCl₃ and / or other salts of iron (III) and copper (II) are widely used as chemical oxidants. The yield and conductivity of the PPy are affected by a variety of factors such as solvent and oxidant, initial pyrrole/oxidant ratio, duration and temperature of the reaction (Vernitskaya & Efimov, 1997; Cho *et al.*, 2007). Figure 1.3 shows the schematic preparation of the PPy.

n
$$N$$

H $Fe^{3+} \rightarrow Fe^{2+}$ H

Figure 1.3 The oxidative polymerization of PPy.

1.5.3 Application of conducting polymers

The conducting polymers have found much interesting application in various areas such as energy storage devices (Yoneyama *et al.*, 1991), electrocatalysts, biosensors, photoluminescent and electroluminescent materials (Komaba *et al.*, 1998), artificial muscles, gas separating membranes, anticorrosive coatings, electromagnetic shields, materials for microlithography, electrophotography, etc (Vernitskaya & Efimov, 1997; Komaba *et al.*, 1998; Cho *et al.*, 2007). PPy is intensively colored (black) in the oxidized form while its reducing form shows transparent green that can be utilized in electrochromic switching application (Partch *et al.*, 1991). Soluble polypyrrole as the transparent anode can also be used in polymer light-emitting diodes (Gao *et al.*, 1996). Interestingly, polypyrrole actuator with a bioadhesive surface has been employed for accumulating bacteria from physiological media (Liu *et al.*, 2009).

1.5.4 Conducting polymer-inorganic nanoparticle composites

Conducting polymers are infusible in nature and generally insoluble in the usual solvents of which seriously limit its processability. Therefore, one method of overcoming the processability problem is to prepare the conducting polymers in colloidal form via dispersion *in situ* polymerization with deposition of polymer onto inorganic particle surface (Filocamo & Grinstaff, 2004; Cho *et al.*, 2007). Therefore,

conducting polymer-based inorganic composites seem to find more potential application compared to pure conducting polymers.

1.6 Problem Statements

Nanosilica, owing to excellent mechanical and optical properties, has potential application as a filler to polymeric system in the preparation of nanocomposites. The main disadvantages of the incorporation of the fillers to polymers are agglomeration of particles and formation of non-selective voids at the interface of the particles and the polymer matrix. Therefore, the nature of the interface between the organic and inorganic parts plays a crucial role. By introducing coupling agents or reactive groups onto the surface of nanoparticles, a good interfacial adhesion between inorganic and organic components can also be achieved.

Silica nanoparticle can be prepared via sol-gel process which is very sensitive towards the experimental conditions that affect the rate of hydrolysis and condensation reactions, e.g., the type and concentration of starting materials (alkoxides), H₂O/alkoxide ratio, catalyst nature and concentration/pH, nature of solvent, temperature, time of reaction, aging and drying method (Iller, 1979; Chu *et al.*, 1997; Meixner & Dyer, 1999; Colomer & Anderson, 2001; Enomoto *et al.*, 2005). There are several drawbacks in sol-gel process such as difficulty in controlling the particle size, aggregation and agglomeration, shrinkage and cracking, and reaction time. Some key parameters in the preparation of nanosilicas, such as, effect of feed rate, mixing mode and drying process have been expected to affect on the particle size as well as distribution and aggregation and agglomeration, shrinkage and cracking of the silica nanoparticles. For modification of the surface of silica nanoparticles, two main methods are generally employed; post-modification and *in-situ* modification (co-condensation method) (Hoffmann *et al.*, 2006). Most of the literatures report the application of co-condensation method in the preparation and modification of mesoporous silica by using surfactant or polymeric compound as a template (Fowler *et al.*, 1998; Richer, 1998; Igarashi *et al.*, 1999; Macquarrie *et al.*, 1999; Macquarrie *et al.*, 2001; Etienne *et al.*, 2002; N. Liu *et al.*, 2003; Sadasivan *et al.*, 2003; Gibaud *et al.*, 2004; Shah *et al.*, 2004; Park & Ha, 2006). There are a few works on the preparation of nanoparticles via co-condensation method (Vandenberg *et al.*, 1991; Branda *et al.*, 2007). Although post-modification is commonly used for modification purpose but, there are some drawbacks in this method such as reduction of porosity by attaching organic group to the pore surface and wall, and blocking via diffusion of modifier into the pore for mesoporous silica (Hoffmann *et al.*, 2006), multiple reaction, time and energy consuming as well as the use of organic solvent (e.g. toluene) for silica nanoparticles (Avella *et al.*, 2006; Rahman *et al.*, 2007).

On the other hand, polypyrrole (PPy) as a promising conducting polymer has been widely found of particular interest, owing to their high polarizability, superior conductivity, stability in the oxidised state and interesting redox properties. Although simplicity of the synthetic procedures and availability of the initial monomers are attractive features of PPy, but, bulk polypyrrole is infusible, intractable and insoluble in common solvents leading to a serious limit on its processability (Perruchot *et al.*, 1998). One method of overcoming the processability problem is to prepare conducting polymers in colloidal form via oxidative dispersion *in-situ* polymerization with deposition of polymer onto inorganic or organic particle surface. In many of cases, surfactant and water soluble polymeric overlayer are used for better dispersion of nanocomposite particles.

1.7 Research Objectives

The main objectives of this research are proposed as follows;

- (a) To develop a modified sol-gel method in preparation of silica nanoparticles with controlling particle size, monodispersity, aggregation and agglomeration by focusing on feed rate of ammonia, order of mixing of starting material and drying process.
- (b) To develop an organo-modification method for synthesizing of modified silica nanoparticles via *in-situ* modification (co-condensation method) by emphasizing on mass loading of modifier (organosilanes), solvent type and optical properties.
- (c) To compare between *in-situ* and post-modification process by emphasizing on mass loading of modifier, physical properties (density, specific surface area, and porosity), morphological properties, optical properties and studying thin film preparation of modified silica.
- (d) To establish a method in order to synthesize core-shell composite system in nanoscale with silica and modified silica nanoparticle as a core and polypyrrole as a conducting polymer shell by emphasizing on preparation method, pre-deposition mode of monomer (Py) or oxidant (FeCl₃.6H₂O and / or (NH₄)₂S₂O₈), pre-deposition time, mass loading of PPy on the surface of substrate (silica and modified silica), oxidant type and dopant type.

1.8 Scope of the Study

This study investigates the synthesis of silica nanoparticles and their potential applications in the preparation of inorganic-organic conducting polymer nanocomposite particles using pre-chemical modification of silica surface and bulk. This research is divided into three parts: (i) synthesis of silica nanoparticles by emphasizing on the controlling parameters on aggregation and agglomeration, (ii) surface modification of silica by adopting two methods *in-situ* and post-modification technique by emphasizing the amount of modifier mass loading that affects on aggregation and agglomeration of particulate system and (iii) synthesis of core-shell system using silica and modified silica-PPy nanocomposite particles with expectation towards lower aggregation, higher monodispersity and higher conductivity.

In the first part, some effective parameters on particle size and dispersity with focusing on aggregation and agglomeration are described. Three key factors such as, effect of feed rate, mixing mode and drying process are used to control and reduce the undesirable phenomena, i.e. aggregation and agglomeration (feed rate and mixing mode), the control of particle size (feed rate and mixing mode), and shrinkage and cracking (drying process).

In the second part, co-condensation method is introduced in the preparation of organic-inorganic hybrid materials via *in-situ* chemical modification of silica particle. The effect of mass loading of modifier on particle size and distribution and gelling time of reaction is investigated. When methanol is used instead of ethanol as a solvent a significant effect on particle size and distribution can be expected. Photoluminescence, solid state UV and Raman spectroscopy are utilized in order to

18

study optical properties of *in-situ* modified silica in comparison with that in pure silica. In addition, a comparison study between *in-situ* modified and post-modified silica (prepared in non-aqueous and aqueous media) is done by emphasizing the organic content on the surface in which the dispersion of particle could be improved by reducing the aggregation and agglomeration. Specific surface area, porosity, density, morphological and optical properties, and thin film coating on glass slide are obtained in order to study comparative aspects.

In the last part, all particles prepared in former parts, silica, *in-situ* modified and post-modified (in non-aqueous and aqueous solvent) are used as substrate or nano-core for chemical deposition of polypyrrole (as nano-shell) in the preparation of core-shell nanocomposite systems. The effect of preparation method and mode of pre-deposition of monomer (pyrrole) or oxidant, deposition time, type of oxidant (FeCl₃.6H₂O and (NH₄)₂S₂O₈), type of dopant (*p*-tosylate) on conductivity, particle size and distribution are studied.

1.9 Thesis layout

This thesis contains of 7 chapters. Chapter 1 provides a brief introduction to general aspects of silica nanoparticles, modification processes, conducting polymers, coreshell nanocomposites and objectives of this study. Chapter 2 consists of literature reviews, mainly on the synthesis of silica nanoparticles via sol-gel process, chemical modification of silica via co-condensation and post-modification method, and the synthesis of PPy and silica-PPy nanocomposites via dispersion polymerization. The materials, instruments, experimental procedures, and characterization techniques used in this research are described in Chapter 3. Chapters 4 to 6 are actually the results and discussion chapters according to the research objectives. Chapter 4 describes the synthesis of silica nanoparticles via sol-gel process. Chapter 5 introduces the synthesis of organo-functionalized silica nanoparticle by co-condensation method with further study on optical properties. In addition, this chapter investigates a comparative study of *in-situ* and post-modification process based on physical, morphological, and optical properties. Chapter 6 demonstrates the preparation and characterization of silica and modified silica-PPy nanocomposite particle via oxidative dispersion polymerization. Finally, the thesis ends with the conclusion and recommendation for future work in Chapter 7.

REFERENCES

- Aguiar, A., Gonzlez-Villegas, S., Rabelero, M., Mendizbal, E., Puig, J. E., Domnguez, J. M., & Katime, I. (1999). Core-shell polymers with improved mechanical properties prepared by microemulsion polymerization. *Macromolecules*, 32(20), 6767-6771.
- Ajayan, P. M., Schadler, L. S., & Braun, P. V. (2003). Nanocomposite science and technology. Weinheim: Wiley-VCH.
- Asefa, T., MacLachlan, M. J., Coombs, N., & Ozin, G. A. (1999). Periodic mesoporous organosilicas with organic groups inside the channel walls. *Nature*, 402(6764), 867-871.
- Aurobind, S. V., Amirthalingam, K. P., & Gomathi, H. (2006). Sol-gel based surface modification of electrodes for electro analysis. *Adv. Colloid Interface Sci.*, 121(1-3), 1-7.
- Avella, M., Bondioli, F., Cannillo, V., Pace, E. D., Errico, M. E., Ferrari, A. M., Focher, B., & Malinconico, M. (2006). Poly([epsilon]-caprolactone)-based nanocomposites: Influence of compatibilization on properties of poly([epsilon]-caprolactone)-silica nanocomposites. *Compos. Sci. Technol.*, 66(7-8), 886-894.
- Basinska, T. (2005). Hydrophilic core-shell microspheres: A suitable support for controlled attachment of proteins and biomedical diagnostics. *Macromol. Biosci.*, 5(12), 1145-1168.

- Bergna, H. E. (2006). Silica supported catalysts and method of preparation. In H. E. Bergna & W. O. Roberts (Eds.), *Colloidal silica: Fundamentals and applications* (pp. 807-814). Boca Raton: CRC Press.
- Branda, F., Silvestri, B., Luciani, G., & Costantini, A. (2007). The effect of mixing alkoxides on the Stöber particles size. *Colloids Surf. A*, 299(1-3), 252-255.
- Brinker, C. J. (1988). Hydrolysis and condensation of silicates: Effects on structure. *J. Non-Cryst. Solids*, 100(1-3), 31-50.
- Brinker, C. J. (2006). Sol-gel processing of silica. In H. E. Bergna & W. O. Roberts (Eds.), *Colloidal silica: Fundamentals and applications* (pp. 615-635). Boca Raton: CRC Press.
- Brinker, C. J., & Scherer, G. W. (1990). Sol-gel science: The physics and chemistry of sol-gel processing. San Diego: Academic Press.
- Brinkmann, U., Ettlinger, M., Kerner, D., & Schmoll, R. (2006). Synthetic amorphous silicas. In H. E. Bergna & W. O. Roberts (Eds.), *Colloidal silica: Fundamentals and applications* (pp. 575-588). Boca Raton: CRC Press.
- Cao, W., & Hench, L. L. (1996). Bioactive materials. Ceram. Int., 22(6), 493-507.
- Carswell, A. D. W., & Grady, B. P. (2004). Electrically conducting polymeric nanostructures: Techniques for one-dimensional synthesis. In J. A. Schwarz, C. I. Contescu & K. Putyera (Eds.), *Dekker encyclopedia of nanoscience and nanotechnology* (Vol. 2, pp. 1015-1023). New York: Marcel Dekker.
- Caruso, F. (2001). Nanoengineering of particle surfaces. Adv. Mater., 13(1), 11-22.
- Celzard, A., & Marêché, J. F. (2002). Application of the sol-gel process using well-tested recipes. J. Chem. Edu., 79(7), 854-859.
- Chen, Y., Zhou, S., Yang, H., & Wu, L. (2005). Structure and properties of polyurethane/nanosilica composites. J. Appl. Polym. Sci., 95(5), 1032-1039.
- Cho, S. H., Song, K. T., & Lee, J. Y. (2007). Recent advances in polypyrrole. In T. A. Skotheim & J. R. Reynolds (Eds.), *Conjuagated polymers: Theory, synthesis, properties, and characterization.* Boca Raton: CRC Press.
- Chu, L., Tejedor-Tejedor, M. I., & Anderson, M. A. (1997). Particulate sol-gel route for microporous silica gels. *Microporous Mater.*, 8(5-6), 207-213.
- Colomer, M. T., & Anderson, M. A. (2001). High porosity silica xerogels prepared by a particulate sol-gel route: pore structure and proton conductivity. *J. Non-Cryst. Solids*, 290(2-3), 93-104.

- Coltrain, B. K., & Kelts, L. W. (2006). The chemistry of hydrolysis and condensation of silica sol-gel precursors. In H. E. Bergna & W. O. Roberts (Eds.), *Colloidal silica: Fundamentals and applications* (pp. 637-644). Boca Raton: CRC Press.
- Enomoto, N., Kumagai, A., & Hojo, J. (2005). Aging effect of starting solutions for spherical silica synthesis. J. Ceram. Soc. Jpn., 113(5), 340-343.
- Etienne, M., Lebeau, B., & Walcarius, A. (2002). Organically-modified mesoporous silica spheres with MCM-41 architecture. *New J. Chem.*, *26*(4), 384-386.
- Falcone, J. S. (2006). Silica in biology. In H. E. Bergna & W. O. Roberts (Eds.), Colloidal silica: Fundamentals and applications (pp. 765-778). Boca Raton: CRC Press.
- Filocamo, S. F., & Grinstaff, M. W. (2004). Nanostructures based on conducting polymers. In J. A. Schwarz, C. I. Contescu & K. Putyera (Eds.), *Dekker encyclopedia of nanoscience and nanotechnology* (Vol. 3, pp. 2615-2626). New York: Marcel Dekker.
- Fowler, C. E., Lebeau, B., & Mann, S. (1998). Covalent coupling of an organic chromophore into functionalized MCM-41 mesophases by template-directed co-condensation. *Chem. Commun.*, (17), 1825-1826.
- Gao, J., Heeger, A. J., Lee, J. Y., & Kim, C. Y. (1996). Soluble polypyrrole as the transparent anode in polymer light-emitting diodes. *Synth. Met.*, 82(3), 221-223.
- Gibaud, A., Bardeau, J.-F., Dutreilh-Colas, M., Bellour, M., Balasubramanian, V. V., Robert, A., Mehdi, A., Reyé, C., & Corriu, R. J. (2004). Influence of functional organic groups on the structure of CTAB templated organosilica thin films. J. Mater. Chem., 14(12), 1854-1860.
- Gomes, D., Nunes, S. P., & Peinemann, K.-V. (2005). Membranes for gas separation based on poly(1-trimethylsilyl-1-propyne)-silica nanocomposites. *J. Membr. Sci.*, 246(1), 13-25.
- Gong, X., Peng, S., Wen, W., Sheng, P., & Li, W. (2009). Design and fabrication of magnetically functionalized core/shell microspheres for smart drug delivery. *Adv. Funct. Mater.*, 19(2), 292-297.
- Haag, R. (2004). Supramolecular drug-delivery systems based on polymeric coreshell architectures. *Angew. Chem. Int. Ed.*, 43(3), 278 -282.
- Hench, L. L. (1997). Sol-gel materials for bioceramic applications. *Curr. Opin. Solid State Mater. Sci.*, 2(5), 604-610.
- Hench, L. L. (1998). Sol-gel silica: Properties, processing and technology transfer. New Jersey: Noyes Publication.

Hench, L. L., & West, J. K. (1990). The sol-gel process. Chem. Rev., 90(1), 33-72.

- Hoffmann, F., Cornelius, M., Morell, J., & Froba, M. (2006). Silica-based mesoporous organic-inorganic hybrid materials. *Angew. Chem. Int. Ed.*, 45(20), 3216-3251.
- Igarashi, N., Tanaka, Y., Nakata, S.-I., & Tatsumi, T. (1999). Increased stability of organically modified MCM-41 synthesized by a one-step procedure. *Chem. Lett.*, 28(1), 1-2.
- Iller, R. K. (1979). The chemistry of silica. New York: John Wiley & sons.
- Inagaki, S., Guan, S., Fukushima, Y., Ohsuna, T., & Terasaki, O. (1999). Novel mesoporous materials with a uniform distribution of organic groups and inorganic oxide in their frameworks. J. Am. Chem. Soc., 121(41), 9611-9614.
- Kashiwagi, T., Morgan, A. B., Antonucci, J. M., VanLandingham, M. R., Harris, R. H., Awad, W. H., & Shields, J. R. (2003). Thermal and flammability properties of a silica-poly(methylmethacrylate) nanocomposite. J. Appl. Polym. Sci., 89(8), 2072-2078.
- Klein, L. C. (1988). Sol-gel technology for thin films, fibers, preforms, electronics, and speciality shapes. New Jersey: Noyes Publication.
- Köhler, M., & Fritzsche, W. (2004). Nanotechnology: An introduction to nanostructuring techniques. Weinheim: Wiley-VCH.
- Komaba, S., Fujihana, K., Osaka, T., Aiki, S., & Nakamura, S. (1998). Electrochemical formation of polypyrrole/SiO₂ composite film and its application to organic electroluminescence devices. J. Electrochem. Soc., 145(4), 1126-1130.
- Koo, J. H. (2006). Polymer nanocomposites. New York: McGraw-Hill.
- Lev, O., Wu, Z., Bharathi, S., Glezer, V., Modestov, A., Gun, J., Rabinovich, L., & Sampath, S. (1997). Sol-gel materials in electrochemistry. *Chem. Mater.*, *9*(11), 2354-2375.
- Li, W.-H., & Stöver, H. D. H. (2000). Monodisperse cross-linked core-shell polymer microspheres by precipitation polymerization. *Macromolecules*, *33*(12), 4354-4360.
- Liu, A., Zhao, L., Bai, H., Zhao, H., Xing, X., & Shi, G. (2009). Polypyrrole actuator with a bioadhesive surface for accumulating bacteria from physiological media. ACS Appl. Mater. Interfaces, 1(4), 951-955.
- Liu, N., Assink, R. A., Smarsly, B., & Brinker, C. J. (2003). Synthesis and characterization of highly ordered functional mesoporous silica thin films with positively chargeable-NH₂ groups. *Chem. Commun.*, (10), 1146-1147.

- Liu, Y.-L., Hsu, C.-Y., Wei, W.-L., & Jeng, R.-J. (2003). Preparation and thermal properties of epoxy-silica nanocomposites from nanoscale colloidal silica. *Polymer*, 44(18), 5159-5167.
- Macquarrie, D. J., Jackson, D. B., Mdoe, J. E. G., & Clark, J. H. (1999). Organomodified hexagonal mesoporous silicates. *New J. Chem.*, 23(5), 539-544.
- Macquarrie, D. J., Jackson, D. B., Tailland, S., & Utting, K. A. (2001). Organically modified hexagonal mesoporous silicas (HMS)-remarkable effect of preparation solvent on physical and chemical properties. J. Mater. Chem., 11(7), 1843-1849.
- Meixner, D. L., & Dyer, P. N. (1999). Influence of sol-gel synthesis parameters on the microstructure of particulate silica xerogels. J. Sol-Gel Sci. Technol. 14(3), 223-232.
- Melde, B. J., Holland, B. T., Blanford, C. F., & Stein, A. (1999). Mesoporous sieves with unified hybrid inorganic/organic frameworks. *Chem. Mater.*, 11(11), 3302-3308.
- Muromachi, T., Tsujino, T., Kamitani, K., & Maeda, K. (2006). Application of functional coating by sol-gel method. J. Sol-Gel Sci. Technol., 40(2-3), 267-272.
- Nabok, A. (2005). Organic and inorganic nanostructures. Boston: ARTECH House.
- Parida, S. K., Dash, S., Patel, S., & Mishra, B. K. (2006). Adsorption of organic molecules on silica surface. Adv. Colloid Interface Sci., 121(1-3), 77-110.
- Park, S. S., & Ha, C.-S. (2006). Organic-inorganic hybrid mesoporous silicas: Functionalization, pore size, and morphology control. *Chem. Record*, 6(1), 32-42.
- Partch, R., Gangolli, S. G., Matijevic, E., Cal, W., & Arajs, S. (1991). Conducting polymer composites: I. Surface-induced polymerization of pyrrole on iron(III) and cerium(IV) oxide particles. J. Colloid Interface Sci., 144(1), 27-35.
- Perruchot, C., Chehimi, M. M., Mordenti, D., Briand, M., & Delamar, M. (1998). The role of a silane coupling agent in the synthesis of hybrid polypyrrolesilica gel conducting particles. J. Mater. Chem., 8(10), 2185-2193.
- Poole, P. C., & Owens, F. J. (2003). *Introduction to nanotechnology*. New Jersey: John Wiley & Sons.
- Qi, D.-M., Bao, Y.-Z., Weng, Z.-X., & Huang, Z.-M. (2006). Preparation of acrylate polymer/silica nanocomposite particles with high silica encapsulation efficiency via miniemulsion polymerization. *Polymer*, 47(13), 4622-4629.

- Rahman, I. A., Vejayakumaran, P., Sipaut, C. S., Ismail, J., Bakar, M. A., Adnan, R., & Chee, C. K. (2007). An optimized sol-gel synthesis of stable primary equivalent silica particles. *Colloids Surf. A*, 294(1-3), 102-110.
- Rao, C. N. R., & Cheetham, A. K. (2006). Material science at the nanoscale. In Y. Gogotsi (Ed.), *Nanomaterials handbook*. Boca Raton: CRC Press.
- Rao, C. N. R., Müller, A., & Cheetham, A. K. (2004). Nanomaterials: An introduction. In C. N. R. Rao, A. Müller & A. K. Cheetham (Eds.), *The chemistry of nanomaterials* (pp. 1-11). Weinheim: Wiley-VCH.
- Reisfeld, R., & Saraidarov, T. (2006). Innovative materials based on sol-gel technology. *Opt. Mater.*, 28(1-2), 64-70.
- Richer, R. (1998). Direct synthesis of functionalized mesoporous silica by non-ionic alkylpolyethylene oxide surfactant assembly. *Chem. Commun.*, (16), 1775-1776.
- Sadasivan, S., Khushalani, D., & Mann, S. (2003). Synthesis and shape modification of organo-functionalized silica nanoparticles with ordered mesostructured interiors. *J. Mater. Chem.*, *13*(5), 1023-1029.
- Sarid, D., McCarthy, B., & Jabbour, G. E. (2004). Nanotechnology for data storage application. In B. Bhushan (Ed.), *Springer handbook of nanotechnology* (pp. 899-920). Heidelberg: Springer-Verlag.
- Schärtl, W. (2000). Crosslinked spherical nanoparticles with core-shell topology. *Adv. Mater.*, *12*(24), 1899-1908.
- Schmid, G. (2004). *Nanoparticles: From theory to applications*. Weinheim: Wiley-VCH.
- See, K. H., Mullins, M. E., Mills, O. P., & Heiden, P. A. (2005). Core-shell, sol-gel: A reactive core-shell nanoparticle approach to prepare hybrid nanocomposites: effects of processing variables. *Nanotechnology*, 16(9), 1950-1959.
- Seo, J. W., & Kim, B. K. (2005). Preparations and properties of waterborne polyurethane / nanosilica composites. *Polym. Bull.*, 54(1-2), 123-128.
- Shah, J., Kim, S.-S., & Pinnavaia, T. J. (2004). A versatile pathway for the direct assembly of organo-functional mesostructures from sodium silicate. *Chem. Commun.*, (5), 572-573.
- Sharma, P. K., Miao, W., Giri, A., & Raghunathan, S. (2004). Nanomaterials: Manufacturing, processing, and applications. In J. A. Schwarz, C. I. Contescu & K. Putyera (Eds.), *Dekker encyclopedia of nanoscience and nanotechnology* (Vol. 3, pp. 2435-2450). New York: Marcel Dekker.